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(54) **OPTOELECTRONIC DEVICE AND METHOD FOR PRODUCING AN OPTOELECTRONIC DEVICE**

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See application file for complete search history.

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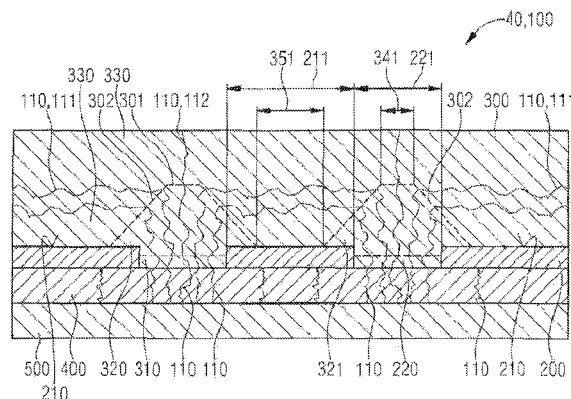
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(57) **ABSTRACT**

A method is provided for producing an optoelectronic device, comprising the steps of providing a substrate, applying a nucleation layer on a surface of the substrate, applying and patterning a mask layer on the nucleation layer, growing a nitride semiconductor in a first growth step, wherein webs are laid which form a lateral lattice, wherein the webs have trapezoidal cross-sectional areas in places in the direction of growth, and laterally overgrowing the webs with a nitride semiconductor in a second growth step, to close spaces between the webs.

14 Claims, 4 Drawing Sheets



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FIG 1

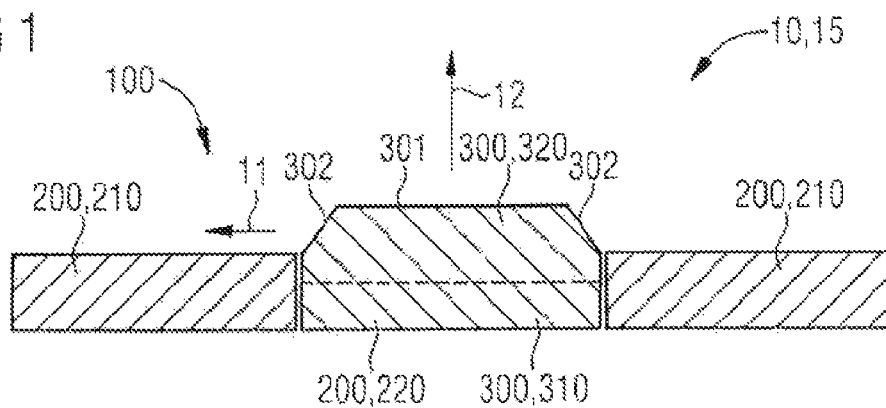


FIG 2

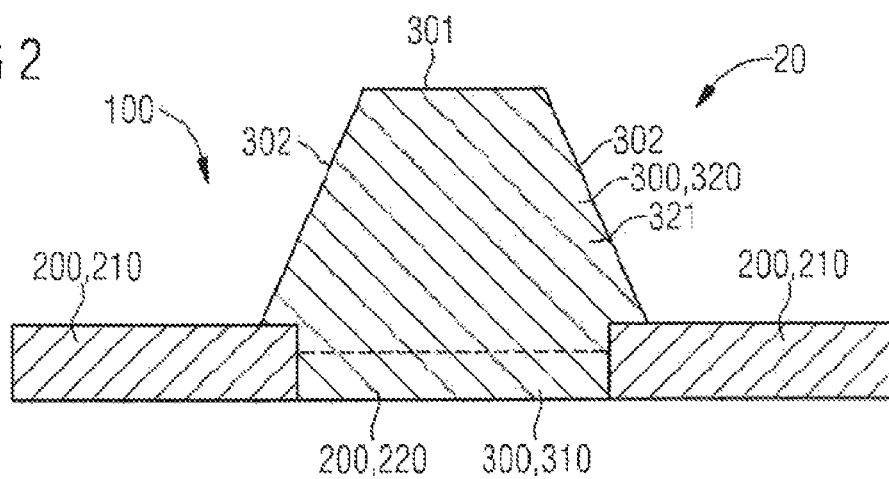


FIG 3

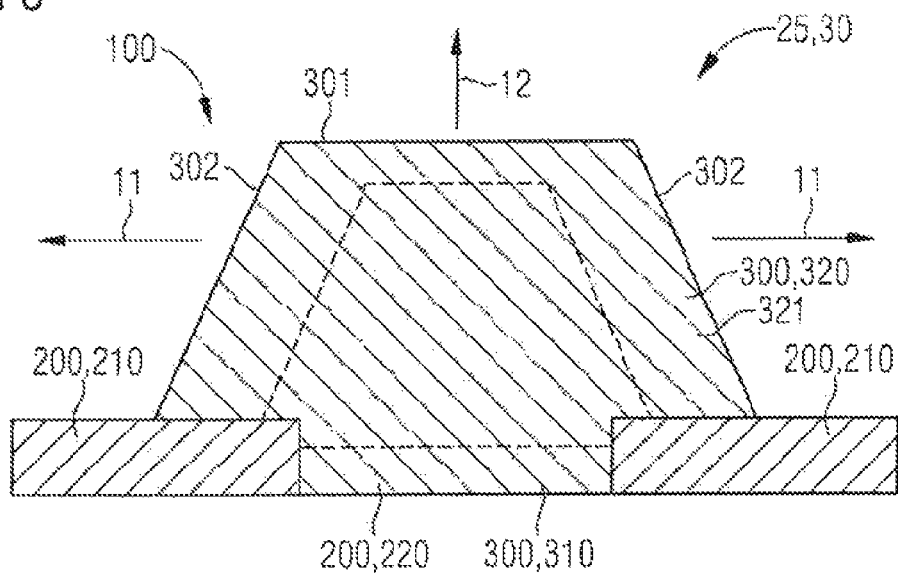


FIG 4

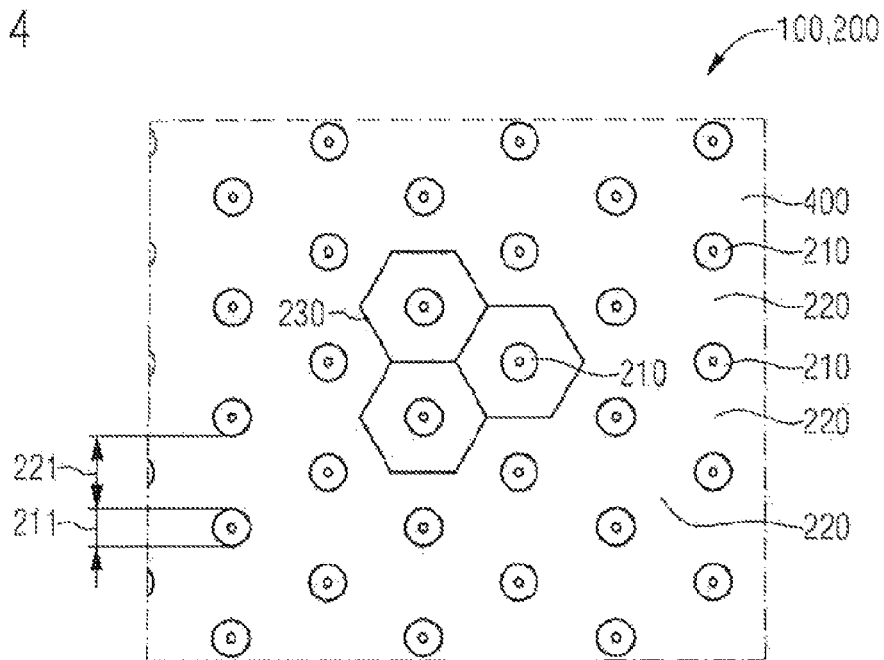


FIG 5

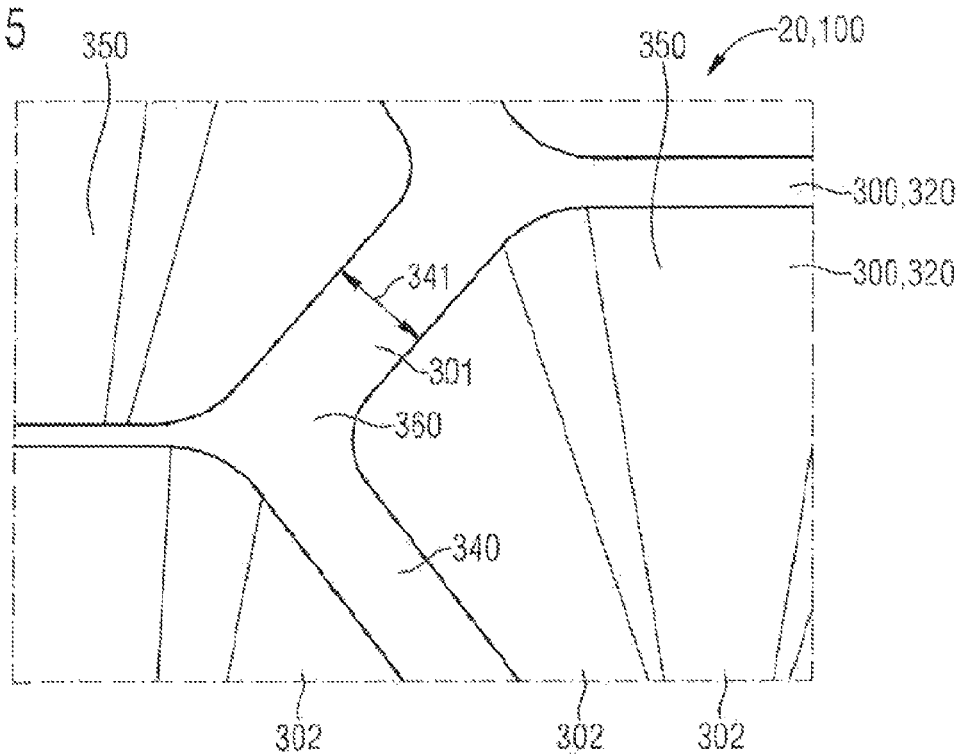


FIG 6

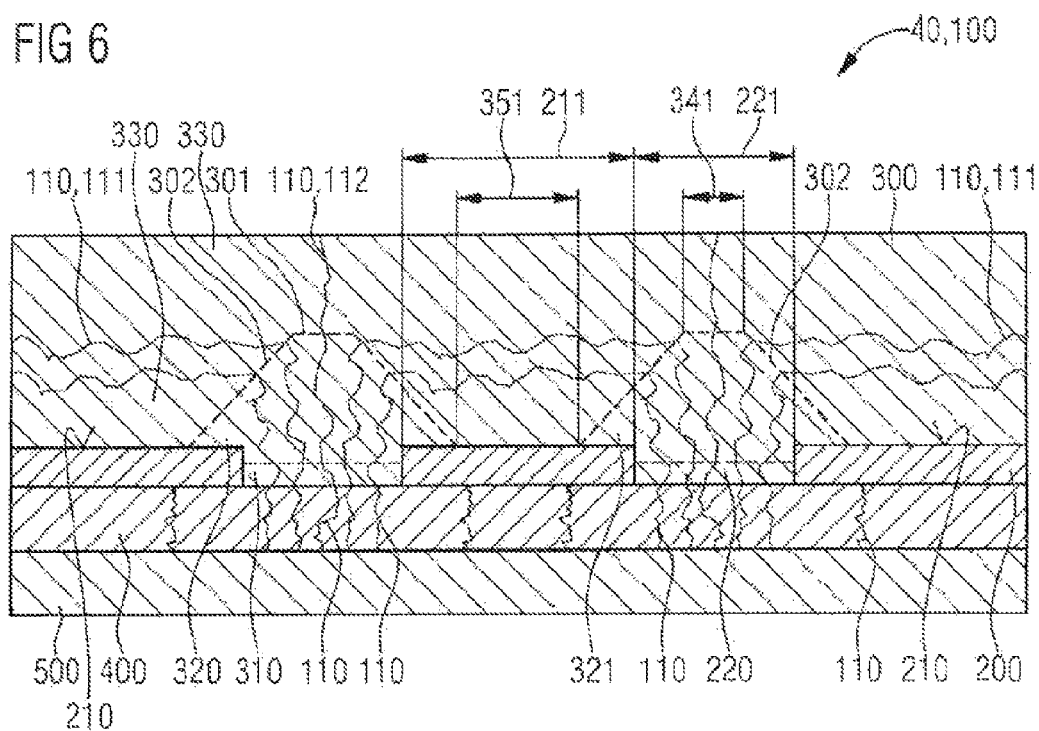


FIG 7

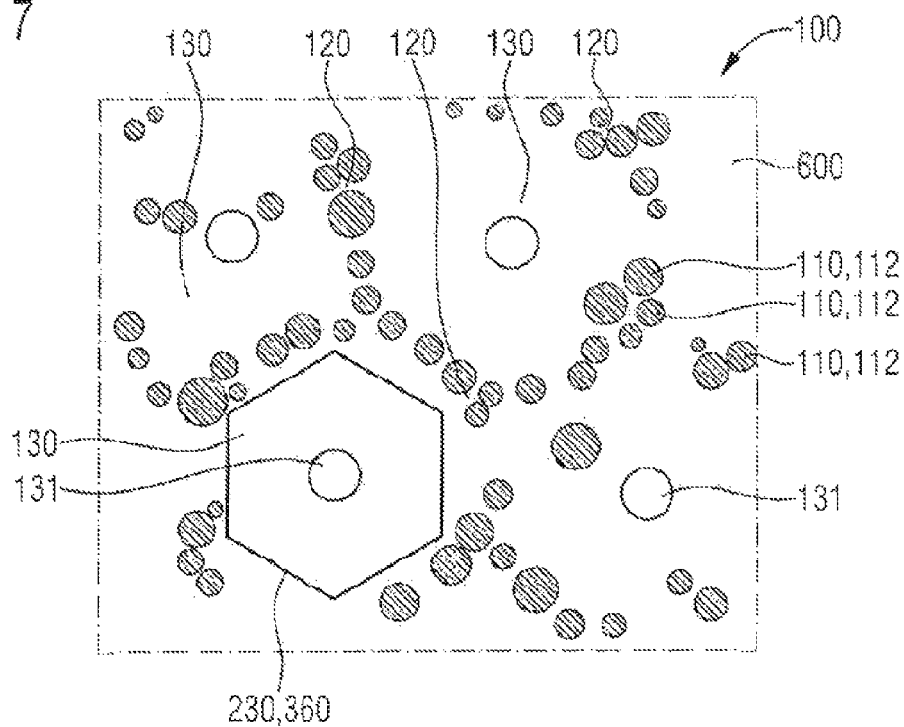
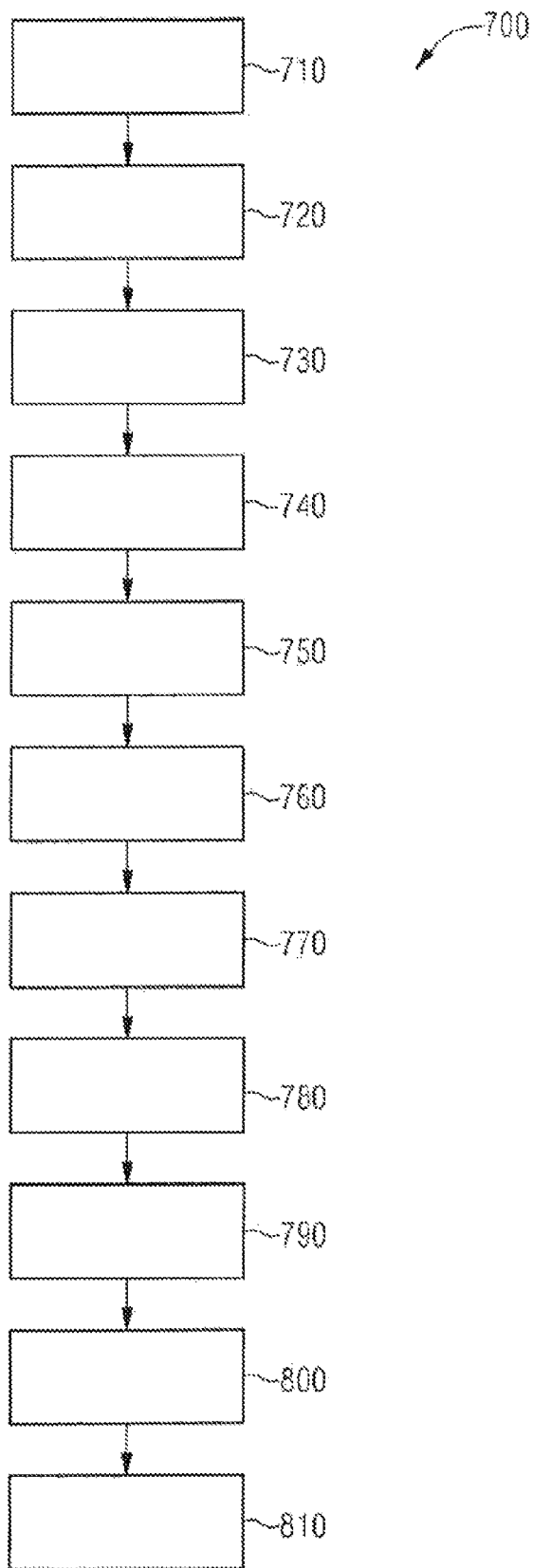


FIG 8



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OPTOELECTRONIC DEVICE AND METHOD FOR PRODUCING AN OPTOELECTRONIC DEVICE

The present invention relates to a method for producing an optoelectronic device and to an optoelectronic device.

In the case of light-emitting diodes based on aluminium indium gallium nitride (AlInGaN), it is known that an emitted light output grows in a less than linear manner with the input electrical power as said electrical power increases. This phenomenon is known as droop.

It is known that the linearity of such light-emitting diodes may be increased through the use of active zones with a plurality of quantum films. Conventionally up to 30 quantum films are provided. (Al)(In)GaN barriers are arranged between the individual quantum films, in order to keep constant the crystal quality in the quantum film sequence. The barriers typically have a thickness of between 3 nm and 5 nm. Maximally thin barriers are desirable, since they form lower piezoelectric barriers for charge carriers moving in the vertical direction and thus allow improved vertical charge carrier transport. It is known that the barriers may be thinner, the lower the dislocation density in the crystal in the plane of the active zone. The dislocation density is typically in the range between $20 \times 10^7 \text{ cm}^{-2}$ and $100 \times 10^7 \text{ cm}^{-2}$.

One object of the present invention consists in providing an improved method for producing an optoelectronic device. A further object of the present invention consists in providing an improved optoelectronic device. These objects are achieved by a method or an optoelectronic device having the features of the independent claims. Preferred further developments are stated in the dependent claims.

A method for producing an optoelectronic device comprises steps of providing a substrate, applying a nucleation layer on a surface of the substrate, applying and patterning a mask layer on the nucleation layer, growing a nitride semiconductor in a first growth step, wherein webs are laid which form a lateral lattice, wherein the webs have trapezoidal cross-sectional areas in places in the direction of growth, and laterally overgrowing the webs with a nitride semiconductor in a second growth step, to close spaces between the webs. Advantageously, this method allows the production of an optoelectronic device with laterally varying defect density. Defect-rich lateral portions of the optoelectronic device may then advantageously carry an increased vertical current flow. Charge carriers from the defect-rich lateral portions may advantageously diffuse into lateral regions with fewer defects and there radiantly recombine.

In one embodiment of the method, the spaces between the webs are of frustopyramidal configuration. Advantageously, the lateral overgrowth of the webs in the second growth step brings about bending of dislocation defects at the facets of the webs facing the frustopyramidal spaces, whereby a reduced defect density forms in regions above the frustopyramidal spaces.

In one embodiment of the method disc-shaped lateral mask regions are applied during patterning of the mask layer. Advantageously, the spaces above the disc-shaped lateral mask regions thereby form during the first growth step.

In one embodiment of the method, the disc-shaped mask regions are arranged in a lateral hexagonal lattice. Advantageously, this results in complete, uniform lateral tiling.

In one embodiment of the method, the disc-shaped mask regions are applied with a diameter of between 0.5 μm and 3 μm . Advantageously, the resultant lateral portions with

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reduced defect density thereby have a suitable size to be supplied by charge carriers diffusing in from adjacent defect-rich portions.

In one embodiment of the method, the disc-shaped mask regions are applied separated from one another by regions with a width of between 0.5 μm and 2 μm . Advantageously, the resultant lateral portions of elevated defect density thereby have a size which allows a favourable compromise between sufficient vertical current flow and the avoidance of non-radiant charge carrier recombination.

In one embodiment of the method, the second growth step is performed under growth parameters which promote coalescence. Advantageously, in the second growth step the spaces between the webs thereby substantially close up, whereby in the regions of the spaces lateral portions with reduced defect density arise.

In one embodiment of the method, the first growth step is performed at a lower temperature than the second growth step. Advantageously, in the first growth step this results substantially in growth in the vertical direction, while during the second growth step growth in the lateral direction primarily takes place.

In one embodiment of the method, after the second growth step a functional layer sequence is deposited, which comprises an actively luminous layer. Advantageously, a larger number of V-defects thereby forms in regions of the actively luminous layer arranged above the defect-rich lateral portions than in the regions above the lateral portions with fewer defects. Advantageously, the higher density of V-defects allows an elevated current flow in the vertical direction.

In one embodiment of the method, a nitride semiconductor is deposited in openings in the mask layer prior to the first growth step, wherein etching back is then carried out. This advantageously results in reduced distortions in the multilayer structure which arises.

In one embodiment of the method, the nucleation layer comprises aluminium (Al). Advantageously, epitaxial growth may be carried out on a nucleation layer comprising aluminium.

In one embodiment of the method, the mask layer comprises two layers which comprise silicon dioxide (SiO_2). The layers of the mask layer comprising silicon dioxide are in this case separated by a layer comprising silicon nitride (SiN). Advantageously, the layers of this mask layer comprising silicon dioxide may then be separated from one another during a subsequent method step at the layer comprising silicon nitride.

In one embodiment of the method, the multilayer structure arranged on the nucleation layer is detached, wherein parts of the mask layer arranged on the detached multilayer structure are used as a hard mask for producing patterned outcoupling structures. Advantageously, no additional mask structures have then to be applied to produce the patterned outcoupling structures. This advantageously simplifies the method.

An optoelectronic device has a multilayer structure which has a higher dislocation density in a lattice-shaped lateral region than in other lateral regions. Advantageously, in this optoelectronic device an elevated vertical current may flow in the lateral regions with higher dislocation density. In this case, flowing charge carriers may advantageously diffuse out of the lattice-shaped lateral region with higher dislocation density into the adjacent other lateral regions and there radiantly recombine. This advantageously results in good high current linearity of the optoelectronic device.

In one embodiment of the optoelectronic device, a gaseous inclusion is arranged in another lateral region surrounded by the lattice-shaped lateral region. Advantageously, the gaseous inclusion points to otherwise complete closure of the lateral region with lower dislocation density.

In one embodiment of the optoelectronic device, the multilayer structure has an actively luminous layer. In this case, the actively luminous layer has a higher density of V-defects in the lattice-shaped lateral region than in the other lateral regions. Advantageously, the V-defects in the lattice-shaped lateral region with higher dislocation density bring about a reduction in the piezoelectric barriers to be overcome by charge carriers flowing in the vertical direction, whereby in the lattice-shaped lateral region an elevated current may flow in the vertical direction through the actively luminous layer of the multilayer structure. Consequently, charge carriers may diffuse out of the lattice-shaped lateral region into the other lateral regions and there radiantly recombine.

In one embodiment of the optoelectronic device, the lattice-shaped lateral region forms a hexagonal lattice. Advantageously, a hexagonal lattice constitutes favourable lateral tiling, whereby the other lateral regions with reduced dislocation density are distributed uniformly between the lattice-shaped lateral region with higher dislocation density.

In one embodiment of the optoelectronic device, the multilayer structure has an actively luminous layer. In this case, the lattice-shaped lateral region is configured to inject a higher number of charge carriers per surface area into the actively luminous layer than the other lateral region. This advantageously results in good high current linearity of the optoelectronic device.

The above-described characteristics, features and advantages of this invention and the manner in which these are achieved will become clearer and more distinctly comprehensible from the following description of the exemplary embodiments, which are explained in greater detail in connection with the drawings. In the figures, in each case in a highly schematic representation

FIG. 1 shows a section through a multilayer structure in a first processing state;

FIG. 2 shows a section through the multilayer structure in a second processing state;

FIG. 3 shows a section through the multilayer structure in a third processing state;

FIG. 4 is a plan view onto a mask layer of the multilayer structure;

FIG. 5 is a perspective representation of a nitride semiconductor layer of the multilayer structure in the second processing state;

FIG. 6 shows a section through the multilayer structure in a fourth processing state;

FIG. 7 is a plan view onto a functional layer sequence of the multilayer structure; and

FIG. 8 shows a flow diagram of a method for producing an optoelectronic device.

FIG. 1 is a highly schematic representation of a section through a part of a multilayer structure **100** in a first processing state **10**. The multilayer structure **100** is part of an optoelectronic device, in particular of an LED chip of a light-emitting diode.

The multilayer structure **100** comprises a patterned mask layer **200** with lateral mask regions **210**, which are separated from one another by openings **220** in the mask. The mask layer **200** may for example comprise silicon dioxide (SiO₂).

The multilayer structure **100** additionally comprises a nitride semiconductor layer **300**. The nitride semiconductor

layer **300** comprises a first layer part **310**, which is arranged in the region of the mask opening **220** between the lateral mask regions **210** of the mask layer **200**. The first layer part **310** may for example comprise gallium nitride (GaN). The first layer part **310** could also be omitted.

The nitride semiconductor layer **300** further comprises a second layer part **320**. In the first processing state **10** shown in FIG. 1, the second layer part **320** is not yet fully present. The first processing state **10** illustrated in FIG. 1 shows the multilayer structure **100** during performance of a first growth step **15** for growing the second layer part **320**. The second layer part **320** is grown epitaxially during the first growth step **15**. The second layer part **320** preferably likewise comprises gallium nitride.

The growth conditions during the first growth step **15** are selected such that growth of the second layer part **320** takes place mainly in a vertical growth direction **12** on a c-surface **301** of the resultant nitride semiconductor layer **300**. Only slight growth occurs in a lateral growth direction **11** on {11 $\bar{2}$ 2} surfaces of the resultant nitride semiconductor layer **300**. In this way, the resultant second layer part **320** tapers in the lateral direction as the vertical direction increases.

FIG. 2 shows a schematic sectional representation of the multilayer structure **100** in a second processing state **20**. The second processing state **20** is achieved after the first growth step **15** shown in FIG. 1 has been carried out for a limited period. In the sectional representation of FIG. 2, the second layer part **320** of the nitride semiconductor layer **300** produced in the first growth step **15** has a trapezoidal cross-sectional area **321** at least in places.

Had the first growth step **15** been continued for longer, growth of the second layer part **320** would have continued in the vertical growth direction **12** until the side faces **302** of the nitride semiconductor layer **300** touched one another and the c-surface **301** of the nitride semiconductor layer **300** disappeared. Then, the resultant second layer part **320** would have had a triangular cross-sectional area in the sectional representation of FIG. 2. However, the first growth step **15** was broken off at an earlier point.

FIG. 3 shows a schematic sectional representation of the multilayer structure **100** in a third processing state **30** during performance of a second growth step **25**. The second growth step **25** is likewise an epitaxial growth step, but performed under different growth conditions from the first growth step **15**. The second growth step **25** is performed under growth conditions which promote coalescence. During the second growth step **25** a third layer part **330** of the nitride semiconductor layer **300** is grown. The third layer part **330** of the nitride semiconductor layer **300** preferably likewise comprises gallium nitride. During the second growth step **25**, growth primarily takes place in the lateral growth direction **11** due to the coalescence-promoting growth conditions. The third layer part **330** is thus mainly deposited on the side faces **302** of the nitride semiconductor layer **300**. Only slight growth takes place in the vertical growth direction **12**. Only a small part of the third layer part **330** is thus deposited on the c-surface **301** of the nitride semiconductor layer **300**.

FIG. 4 shows a plan view onto a part of the mask layer **200** of the multilayer structure **100**. The mask layer **200** is arranged on a nucleation layer **400** which is visible in the regions of the mask openings **220**. The nucleation layer **400** comprises aluminium (Al). The nucleation layer **400** preferably comprises aluminium nitride (AlN).

The lateral mask regions **210** of the mask layer **200** take the form of discs. Preferably, the disc-shaped lateral mask regions **210** have a circular disc shape which may however also be approximated by a polygon. Each disc-shaped lateral

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mask region **210** has a disc diameter **211**. The disc diameter **211** is preferably in the range between 0.5 μm and 3 μm . For example, the disc diameter **211** may amount to 2 μm .

The disc-shaped lateral mask regions **210** are spaced from one another by the mask openings **220**. Two mutually adjacent lateral mask regions **210** here in each case have a disc spacing **221** from one another. The disc spacing **221** preferably amounts to between 0.5 μm and 2 μm . For example, the disc spacing **221** may amount to 1 μm .

The lateral mask regions **210** are arranged in a regular hexagonal lattice **230**. The hexagonal lattice **230** forms a honeycomb pattern. Each lateral mask region **210** is arranged in the centre of a hexagon of the hexagonal lattice **230**. In this way, each lateral disc-shaped mask region **210** (apart from peripherally arranged lateral mask regions **210**) has six nearest neighbours, which are each spaced from the lateral mask region **210** by the disc spacing **221**.

Instead of the hexagonal lattice **230**, a rectangular lattice or a triangular lattice could also be provided, in which the lateral mask regions **210** are arranged.

FIG. 5 shows a perspective representation of the nitride semiconductor layer **300** of the multilayer structure **100** in the second processing state **20**. The second layer part **320** of the nitride semiconductor layer **300** has formed webs **340**, which are arranged above the mask openings **220** in the mask layer **200**. The webs **340** form a lateral lattice **360**, which reproduces the hexagonal lattice **230** of the mask openings **220**. Each web **340** has a web width **341** at its upper end (remote from the mask layer **200**). The web width **341** is preferably less than the disc spacing **221** of the mask layer **200** and may for example amount to 500 nm.

The webs **340** of the second layer part **320** enclose frustopyramidal spaces **350**, which are arranged above the lateral disc-shaped mask regions **210** of the mask layer **200**. The frustopyramidal spaces **350** expand in frustopyramidal manner as their distance from the mask layer **200** increases. Since the webs **340** form the lateral hexagonal lattice **360**, each frustopyramidal space **350** has an approximately hexagonal base area.

In the second growth step **25**, the frustopyramidal spaces **350** are closed or coalesced in the lateral direction starting from the side faces **302** of the webs **340**. Even the c-surfaces **301** of the webs **340** are partly overgrown in the process.

FIG. 6 shows a schematic sectional representation of the multilayer structure **100** in a fourth processing state **40**. The fourth processing state **40** is achieved after completion of the second growth step **25**. In the fourth processing state **40** the frustopyramidal spaces **350** are completely closed by the third layer part **330**.

FIG. 6 additionally shows a substrate **500** not shown in FIGS. 1, 2 and 3 and the nucleation layer **400** not shown in FIGS. 1, 2 and 3 of the multilayer structure **100**. The substrate **500** forms a carrier for the multilayer structure **100** and preferably comprises sapphire (Al_2O_3). Alternatively, the substrate **500** may comprise silicon (Si), silicon carbide (SiC) or gallium nitride (GaN). The substrate **500** preferably takes the form of a wafer.

FIG. 6 additionally shows a plurality of defects **110** in the nucleation layer **400** and the nitride semiconductor layer **300**. The defects **110** may for example be dislocation defects, which arise due to a lattice mismatch between the nitride semiconductor layer **300**, the nucleation layer **400** and the substrate **500**.

The defects **110** continue in the vertical direction through portions of the multilayer structure **100**. Defects **110** formed in the nucleation layer **400** below the lateral mask regions **210** of the mask layer **200** end at the lateral mask regions

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210. However, defects **110** formed in the nucleation layer **400** below the mask openings **220** in the mask layer **200** continue vertically through the first layer part **310** and the second layer part **320** of the nitride semiconductor layer **300**.

Defects **110** which in the process come into contact vertically with one of the side faces **302** of the second layer part **320**, bend there and continue horizontally in the third layer part **330** as bent defects **111**. These bent defects **111** may cancel one another out and then do not reach the top of the multilayer structure **100**. Defects **110** which continue vertically as far as a c-surface **301** of the second layer part **320** are, in contrast, not bent, but rather continue vertically as continuing defects **112** through the third layer part **330** until they reach the top of the multilayer structure **100**.

This results, at the top of the multilayer structure **100**, in a laterally varying density of the defects **110**. In lateral portions above the webs **340** of the second layer part **320** an elevated defect density results. In the other lateral portions, on the other hand, a reduced defect density results.

At the boundary between the mask layer **200** and the nitride semiconductor layer **300**, two adjacent webs **340** of the second layer part **320** have a web spacing **351** from one another. The web spacing **351** may for example amount to approximately 1 μm .

In a subsequent method step, a functional layer sequence **600** comprising an actively luminous layer is deposited onto the top of the multilayer structure **100** shown in FIG. 6. The actively luminous layer of the functional layer sequence **600** comprises one or more quantum films.

It is known that during growth of an actively luminous layer above defects **110**, V-defects (V-pits) preferably form, i.e. the greater the defect density in a lower-lying layer, the more V-defects form. It is likewise known that side facets of such V-defects have a reduced piezoelectricity relative to c-facets. It is likewise known that less indium is incorporated on facets of such V-defects and the growth rate is reduced in these regions. The result is that V-defects have a lower vertical electrical resistance than defect-free c-surfaces of actively luminous layers. However, in V-defects recombination of holes and electrons takes place predominantly non-radiantly.

FIG. 7 shows a plan view onto the top of the functional layer sequence **600** of the multilayer structure **100**, as may for example be photographed using a PL microscope. In this case, crystal defects appear dark. It is clear that the top of the functional layer sequence **600** comprises lateral regions **120** with a high defect density and lateral regions **130** with a low defect density. The lateral regions **120** with a high defect density are located above the hexagonal lattice **230** of the mask openings **220** of the mask layer **200** and the lateral lattice **360** of the webs **340** of the second layer part **320** of the nitride semiconductor layer **300**. The lateral regions with low defect density **130** form the remaining portions of the top of the functional layer **600** of the multilayer structure **100**.

Every or almost every lateral region with a low defect density **130** comprises a gaseous inclusion **131**. This gaseous inclusion **131** is known in the art as a void. Each gaseous inclusion **131** has been formed in the middle of a frustopyramidal space **350** arranged between webs **340**. In the region of the gaseous inclusion **131**, the frustopyramidal space **350** was ultimately closed during the second growth step **25** by the third layer part **330**.

In the lateral regions **120** with high defect density the actively luminous layer of the functional layer sequence **600** has a higher density of V-defects than in the lateral regions **130** with low defect density. Since the lateral regions **120**

with high defect density have a lower vertical resistance due to the presence of the V-defects, an electric current may flow vertically through the functional layer sequence **600** mainly in the lateral regions **120** with high defect density. Due to the lower series resistance in the vertical direction in the regions **120** with high defect density, more quantum films may be supplied with charge carriers. Since the current density per quantum film is thus lower, the droop of a light-emitting diode produced from the multilayer structure **100** may be reduced. Due to the smaller band gap of the quantum films in the lateral regions **130** with low defect density, charge carriers may diffuse from the lateral regions **120** with high defect density into the lateral regions **130** with low defect density. In the lateral regions **130** with low defect density these charge carriers may radiantly recombine. Consequently, the lateral regions **120** with high defect density bring about a current flow in the vertical direction and inject charge carriers into the lateral regions **130** with low defect density, where these may radiantly recombine.

The relative lateral dimensions of the lateral regions **120** with high defect density and the lateral regions **130** with low defect density relative to one another are selected so as to allow a sufficiently great vertical current flow to achieve a favourable high current linearity, without non-radiant recombination dominating in the lateral regions **120** with high defect density.

FIG. 8 shows a summarising schematic flow diagram of a method **700** for producing an optoelectronic device.

In a first method step **710** the substrate **500** is provided. The substrate **500** is preferably provided as a wafer. Then a multiplicity of optoelectronic devices may be produced simultaneously in parallel on the wafer. The substrate may be a sapphire substrate, a silicon substrate, a silicon carbide substrate or a gallium nitride substrate.

In a second method step **720**, the nucleation layer **400** is applied to a surface of the substrate **500**. The nucleation layer **400** may for example comprise aluminium nitride (AlN). The nucleation layer **400** may for example have a thickness of 50 nm. The nucleation layer **400** may be applied for example by sputtering.

In a third method step **730** the mask layer **200** is applied and patterned. The mask layer **200** may for example comprise a layer sequence of silicon dioxide (SiO₂), silicon nitride (SiN) and a further layer of silicon dioxide. The layers of silicon dioxide may for example have a thickness of approximately 100 nm and the layer of silicon nitride a thickness of approximately 50 nm. The mask layer **200** may for example be sputtered onto the nucleation layer **400**.

After application the mask layer **200** is patterned, for example using a photolithographic method. In the process, the lateral mask regions **210** are laid, spaced by the mask openings **220**. The lateral mask regions **210** are preferably laid in the form of circular discs or approximately circular (for example octagonal) discs. The disc diameter **211** of the lateral mask regions **210** may for example amount to between 0.5 µm and 3 µm, preferably approximately 2 µm. The disc spacings **221** between adjacent lateral mask regions **210** may amount for example to between 0.5 µm and 2 µm, preferably approximately 1 µm. Preferably the lateral mask regions **210** are arranged in a hexagonal lattice.

In a fourth method step **740** the first layer part **310** of the nitride semiconductor layer **300** is deposited in the mask openings **220** of the mask layer **200** and then etched back. The first layer part **310** of the nitride semiconductor layer **300** may for example comprise gallium nitride (GaN) and be deposited in a thickness of approximately 90 nm. Deposition of the first layer part **310** may proceed for example in an

installation for metalorganic vapour phase deposition (MOVPE installation). The material of the first layer part **310** is preferably nominally undoped. After deposition of the first layer part **310**, the first layer part **310** is etched back. This may proceed for example in that a TMGa source of the MOVPE installation is closed and silane (SiH₄) is conveyed into the reactor. An N₂/H₂/NH₃ environment then prevails, in which desorption of gallium nitride and growth of silicon nitride (SiN) takes place. The two processes take place concurrently. Etching back may be performed for example for 5 minutes and serves to control crystal distortion. The fourth method step **740** may be omitted in a simplified variant of the method.

In a fifth method step **750**, the second layer part **320** of the nitride semiconductor layer **300** is grown in the first growth step **15**. The second layer part **320** preferably comprises gallium nitride (GaN). The growth conditions are selected such that growth proceeds predominantly in the vertical growth direction **12** and the webs **340** form with trapezoidal cross-sectional areas **321**. Under these growth conditions, a low ratio of less than 1000 may for example prevail between the Group V semiconductor and the Group III semiconductor. The temperature may be less than 1000° C. The silane (SiH₄) feed may then be stopped again.

In a sixth method step **760** the second growth step **25** is carried out, to grow the third layer part **330** of the nitride semiconductor layer **300**. The third layer part **330** preferably likewise comprises gallium nitride (GaN). The second growth step **25** is performed under growth parameters which lead to a preferred growth in the lateral growth direction **11**. In this way, the frustopyramidal spaces **350** between the webs **340** laid in the preceding method step **750** are closed and coalesced. The growth conditions during the second growth step **25** may for example comprise a high ratio between the Group V semiconductor and the Group III semiconductor and a high temperature of more than 1000° C. After the sixth method step **760** the nitride semiconductor layer **300** is consequently present as a closed layer with a laterally modulated density of defects **110**.

In a seventh method step **770** the functional layer sequence **600** is deposited on the surface of the nitride semiconductor layer **300**. The functional layer sequence **600** comprises an actively luminous layer. The functional layer sequence **600** is preferably an LED structure.

In optional further method steps the multilayer structure **100** produced in the preceding method steps may be further processed to yield a complete optoelectronic device. Method steps **780** to **810** demonstrate such a possibility by way of example.

In the eighth method step **780** the multilayer structure **100** is bonded to a carrier substrate. Bonding proceeds such that the functional layer sequence **600** of the multilayer structure **100** faces the carrier substrate. The carrier substrate may for example be a silicon substrate.

In a ninth method step **790** the substrate **500** and the nucleation layer **400** are detached from the other parts of the multilayer structure **100**. Detachment may proceed for example using a process known as laser lift-off. If the nucleation layer **400** comprises aluminium nitride (AlN), separation does not take place until the nitride semiconductor layer **300** comprising gallium nitride (GaN) for example, due to the optical absorption edge of the aluminium nitride (AlN). In the lateral mask regions **210** of the mask layer **200** the middle layer of silicon nitride (SiN) acts as a separation region. Thus one part, of silicon dioxide (SiO₂), of the lateral mask regions **210** of the mask layer **200** remains on the detached nitride semiconductor layer **300**, while another

part, of silicon dioxide (SiO_2), of the lateral mask regions **210** of the mask layer **200** remains on the nucleation layer **400** and the substrate **500**.

The layer of silicon dioxide (SiO_2) remaining on the nitride semiconductor layer **300** may serve in a tenth method step **800** as a hard mask for producing patterned outcoupling structures.

The layer of silicon dioxide (SiO_2) remaining on the nucleation layer **400** and the substrate **500** may be detached in an eleventh method step **810**. The nucleation layer **400** remains on the substrate **500**. The substrate **500** may then be used again to carry out the method **700**.

This patent application claims priority from German patent application 10 2012 217 644.6, the disclosure content of which is hereby included by reference.

The invention has been illustrated and described in greater detail with reference to the preferred exemplary embodiments. Nevertheless the invention is not restricted to the disclosed examples, but rather other variations may be derived therefrom by a person skilled in the art, without going beyond the scope of the invention.

LIST OF REFERENCE SIGNS

10 First processing state
11 Lateral growth direction
12 Vertical growth direction
15 First growth step
20 Second processing state
25 Second growth step
30 Third processing state
40 Fourth processing state
100 Multilayer structure
110 Defect
111 Bent defect
112 Continuing defect
120 Lateral region with high defect density
130 Lateral region with low defect density
131 Hole
200 Mask layer
210 Lateral mask region
211 Disc diameter
220 Mask opening
221 Disc spacing
230 Hexagonal lattice
300 Nitride semiconductor layer
301 c-Surface
302 Side face
310 First layer part
320 Second layer part
321 Trapezoidal cross-sectional area
330 Third layer part
340 Web
341 Web width
350 Pyramidal space
351 Web spacing
360 Lateral lattice
400 Nucleation layer
500 Substrate
600 Functional layer sequence
700 Method
710 Providing the substrate
720 Applying the nucleation layer
730 Applying and patterning the mask layer
740 Applying the first nitride semiconductor layer part and etching back

750 Growing the second nitride semiconductor layer part in the first growth step

760 Growing the third nitride semiconductor layer part in the second growth step

770 Depositing the functional layer sequence

780 Bonding the multilayer structure to a carrier

790 Detaching the multilayer structure

800 Producing patterned outcoupling structures

810 Reusing the substrate

The invention claimed is:

1. A method for producing an optoelectronic device with the following steps:

providing a substrate;

applying a nucleation layer to a surface of the substrate;

applying and patterning a mask layer on the nucleation layer;

growing a nitride semiconductor in a first growth step, wherein webs are laid which form a lateral lattice, wherein the webs have trapezoidal cross-sectional areas in the direction of growth;

laterally overgrowing the webs with a nitride semiconductor in a second growth step, to close spaces between the webs,

wherein a surface of the webs having trapezoidal cross-sectional areas remote from the substrate is a c-surface running parallel to the substrate; and

depositing, after the second growth step, a functional layer sequence comprising an actively luminous layer, wherein defects that extend in a vertical direction to the c-surface of the webs continue in the vertical direction and bring about a formation of v-defects in the functional layer sequence.

2. The method according to claim **1**, wherein the spaces between the webs are frustopyramidal in form.

3. The method according to claim **1**, wherein disc-shaped lateral mask regions are applied during patterning of the mask layer.

4. The method according to claim **3**, wherein the disc-shaped mask regions are arranged in a lateral hexagonal lattice.

5. The method according to claim **3**, wherein the disc-shaped mask regions are applied with a diameter of between $0.5\ \mu\text{m}$ and $3\ \mu\text{m}$.

6. The method according to claim **3**, wherein the disc-shaped mask regions are applied separated from one another by regions with a width of between $0.5\ \mu\text{m}$ and $2\ \mu\text{m}$.

7. The method according to claim **1**, wherein the second growth step is performed under growth parameters which promote coalescence.

8. The method according to claim **1**, wherein the first growth step is performed at a lower temperature than the second growth step.

9. The method according to claim **1**, wherein after the second growth step a functional layer sequence is deposited, which comprises an actively luminous layer.

10. The method according to claim **1**, wherein a nitride semiconductor is deposited in openings in the mask layer prior to the first growth step, wherein etching back is then carried out.

11. The method according to claim **1**, wherein the nucleation layer comprises aluminium.

12. The method according to claim **1**, wherein the mask layer comprises two layers, which comprise silicon dioxide, and wherein the layers of the mask layer comprising silicon dioxide are separated by a layer comprising silicon nitride.

13. The method according to claim **1**, wherein a multilayer structure arranged on the nucleation layer is detached,

and wherein parts of the mask layer arranged on the detached multilayer structure are used as a hard mask for producing patterned outcoupling structures.

14. A method for producing an optoelectronic device comprising:

- providing a substrate;
- applying a nucleation layer to a surface of the substrate;
- applying and patterning a mask layer on the nucleation layer wherein lateral mask regions in the form of discs are laid by the patterning;
- growing a nitride semiconductor in a first growth step, wherein webs are laid which form a lateral lattice, wherein the webs have trapezoidal cross-sectional areas in the direction of growth, and wherein the growth takes place in a vertical direction on a c-surface of the resultant nitride semiconductor, the c-surface being remote from the substrate and running parallel to the substrate;
- laterally overgrowing the webs with a nitride semiconductor in a second growth step, to close spaces between the webs, wherein defects which come into contact vertically with one of the side faces of the webs bend, and wherein defects which extend in the vertical direction to the c-surface of the webs continue in the vertical direction, so that in lateral portions above the webs an elevated defect density results; and
- depositing, after the second growth step, a functional layer sequence comprising an actively luminous layer, wherein the defects that extend in the vertical direction to the c-surface bring about a formation of v-defects in the functional layer sequence.

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